Analysis & Comparison of Old and New Lead Came

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An aging window with lead that is cracked, deformed and nearing structural failure.

History and Discussion

Post-Industrial Revolution lead is a thoroughly modern substance, with a purity and control of composition that medieval glass window artisans could not begin to imagine. In fact, at the time that many of the famous European windows were created, the producers of their lead cames did not possess the ability to determine what other metals were alloyed with the lead, let alone refine the lead to modern standards or produce consistent alloys. Literature searches revealed that analyses of medieval came indicated that the lead of this time contained silver, antimony, copper, tin, etc., in varying amounts.

By the mid-nineteenth century, modern refining processes were developed that enabled the extraction of these extraneous metals from the lead. Unfortunately, removal of the alloying elements resulted in a much weaker came. The unrefined medieval lead was much better at handling the loading imparted by the glass it contained and the wind forces to which it was exposed. Modern "restoration quality"



An intricately leaded, complex figure from a Tiffany window, before restoration.

lead is reported to be based upon analysis of some medieval cames. As the chemical analysis performed in this investigation shows, "restoration" lead contains a higher percentage of elements known to produce solid-solution hardening of lead than the older, late nineteenth- and early twentieth-century lead. While this means that the restoration lead is stronger than the older lead of higher purity, even this lead, and its medieval counterpart, will eventually fail in service.

The reason the lead will eventually fail in service is due to the nature of the substance. Lead is unresponsive to heat treatment and can spontaneously recrystallize at room temperature, making work-hardening for any useful period of time impossible. Due to its low melting temperature, lead is subject to "creep" at the temperatures in which it is normally used. Creep is a slow, plastic (i.e., permanent, doesn't return to its shape once stress is removed) deformation of materials under constant stress, such as lead came supporting glass in a window. This means that the buckling and came cracking exhibited by many



Samples of deteriorated lead came from the Tiffany window at left.

aging nineteenth- and twentieth-century windows is an inherent and unavoidable structural failure of the lead came resulting from the combination of modern refining processes and the nature of lead itself. While medieval and restoration lead will be better at handling service stress due to their different chemical makeup, even these leads will eventually fail in a similar manner.

As our laboratory testing showed, old cames do not solder as effectively as new cames. Therefore, once a window experiences structural lead failure - broken joints, cracked cames, stretched and collapsed sections – the most effective structural solution is complete replacement of the failed lead. This is true regardless of the reason behind the failure, even lack of support bars. As our testing confirmed, methods that retain the cracked and/or sagging lead - flattening, resoldering or adding support bars – will not produce a structurally sound window. There is no "CPR" for lead that has torn, stretched, ruptured or otherwise deformed.



Fig. 1: Cracks (arrows) in 1913 lead (sample A), as received.



Fig. 2: Cracks (arrows) in 1913 joint (sample A), as received.

Some practitioners have expressed concern that replacing the entire lead matrix negatively affects both the aesthetic and the historic value of a window. However, a failed structural support system is in no way comparable to the cosmetic scratches, dents, or finish cracking of desirable antiques. No reputable antique furniture dealer would suggest repairing a splintered, sagging bureau leg with plywood braces and nails to "preserve the history" of such structural damage. This is analogous to the "flatten, resolder and reinforce" method advocated by some.

Releading a window does not alter the artwork of the glass. It merely replaces failed framework. A reputable restorer will seek to preserve the original glass reveal and any special effects by utilizing accurately replicated lead profiles and preserving any irreplaceable appliqués. Glass is a brittle substance and breaks easily. Once a window has buckled, its glass panes are subjected to loads never intended by the original window designer. The cracked and stretched cames can no longer bear their original loads, and these loads are then transferred to the glass. This is a recipe for the destruction of the glass.

In an effort to preserve "authentic" lead, a window owner or repair facility using the flatten technique greatly increases the likelihood of damage to the glass. This appears to be a classic "throw-the-baby-out-with-the-bathwater" response. Releading provides the opportunity to preserve the artwork in a stained glass window by transferring the loads back to a new lead framework. This allows us to view the artistry of that window as originally intended: flat, structurally sound, and with the original glass preserved.

As tensile-strength testing indicates, lead is a weak, low-strength material. Buckling of the lead framework and associated overload cracking of the came walls is typical of structural failure of a load-bearing member. Given the traditional came profiles, this appears to be inevitable, and a window owner will eventually have to decide whether to save and preserve the old lead or to save and preserve the glass panes.

Background

Stained Glass Resources, Inc., of Hampden, Massachusetts, requested that Massachusetts Materials Research, Inc. (MMR), West Boylston, Massachusetts, compare and evaluate lead cames from various sources and document any age-related differences that may exist. A representative of MMR visited the Stained Glass Resources facility to observe the various steps involved in the releading process. During this visit, there was



Fig. 3: Close up of Fig. 2 solder joint crack (arrow).



Fig. 4: New solder joint on new lead (sample E).



Fig. 5: Fracture surface of 1913 lead (sample A). Mag. 12X



Fig. 6: Fracture surface of 1930 lead (sample B). Mag. 12X

Over the last half-century, as much of the stained glass in the United States reached the age of 100 years, the need for restoration has grown by leaps and bounds, as have the questions surrounding it.

What is restoration? Is there a "right" and a "wrong" way to restore stained glass windows? How does it differ, if at all, from preservation, conservation or repair? What are the historically significant and valuable parts of a window? Is there an expected lifespan for lead, and, if so, what is it? Can this lifespan be shortened or lengthened? What part does climate play in a window's overall condition? What are the signs of impending trouble for a leaded stained glass window? Are there some symptoms we can use as a gauge against a time line? Can some treatments actually do more harm than good? While many of us cite our experience base and anecdotal evidence about lead, do we have any scientific data to back it up?

While the client generally grapples with the difficult issues of the historic or sentimental value of a window, we, as the experts in stained glass, are often called upon for a more quantifiable assessment. Because this can have a dramatic effect on the well-being of a window, it is extremely important for us to be as complete and accurate as possible.

The report on which this article is based was commissioned by Stained Glass Resources in an attempt to understand lead deterioration, its causes, its symptoms and some possible remedies. While it cannot answer all of the questions swirling around various stained glass restoration techniques, it provides some sound scientific evidence for choosing one treatment over another. Perhaps publication of this article will not only answer some of the above questions but spark an open and frank discussion on stained glass restoration techniques and practices, using facts and not conjecture.

> Frederick B. Shea, President Stained Glass Resources, Inc.

discussion of common practices other than complete releading, including: cleaning and recementing, resoldering of cracked joints, partial releading, and flattening with impact, applied weight or heat.

As a result of this visit and the information provided, MMR developed a testing and evaluation plan that included the following:

- Binocular microscope examination of the came samples
- Scanning electron microscope (SEM) examination of fracture surfaces present on old cames
- Energy dispersive x-ray spectroscopy (EDS) analysis of fracture surfaces present on old cames

- Comparative tensile strength testing of old and new cames
- Metallurgical analysis of old and new solder joints and old and new cames

These analyses were chosen as the best ways to present any differences noted between the old cames and new cames, and to evaluate these differences with respect to the structural integrity of a window.

The term "old" as used in this report refers to cames produced from the mid-nineteenth to mid-twentieth centuries. Documenting the effects of time, stress, and atmospheric exposure, as well as the quantifiable differences between partial repairs and total lead replacement can help develop a more scientific way to evaluate window conditions. This analysis was a first step toward that goal.

Result of Technical Investigation Visual Examination

Old cames from two windows were chosen for extensive evaluation and comparison with new cames. Resoldered joints from a repair performed in the mid-1970s were chosen for evaluation with respect to a new joint. Resoldering cracks and joints is reportedly a common practice in window repair. Therefore, comparison of the joints produced is key when evaluating the different remedial practices listed previously, especially the effects of repair or partial releading versus complete releading. The windows chosen for testing were provided by Stained Glass Resources, but the actual samples were selected by MMR. These samples are listed in Table I (page 55).

Visual examination of the older cames revealed a multitude of fine cracks extending into the came from its outer edges. Figure 1 (page 53) shows several of these cracks along a one-inch length of Sample A. The older joints revealed widespread cracking as well.

Figures 2 and 3 (page 53) show joint cracks in Sample A. The crack in the lower joint shown in Figure 2 was later examined with EDS analysis. For comparison, Sample E, a new joint on new lead is shown in Figure 4 (page 53). Visible cracks were common to all older samples and not present on new samples. Since the cracking visible with the naked eye is not necessarily the only cracking present, further microscopic examination was performed.

Binocular Microscope Examination

A binocular microscope is a light microscope of the type commonly pic-

tured when the word "microscope" is mentioned. Another term for this piece of equipment is stereo microscope.

This examination was conducted to allow inspection of the subject cames at magnifications up to 50X. Selected cracks were carefully broken open to reveal their fracture surfaces and examined with this method as well.

This examination did not reveal any new information with regard to the came surfaces. The fracture surfaces, however, were obviously different in appearance from the bright, shiny laboratory-created surfaces formed upon exposing the cracks. When a fracture is opened for inspection, metal that was still intact nearby the crack in question produces a new fracture. This is the laboratory-created fracture. While it is not related to the initial crack, it can provide information about the base metal to compare with the crack in question.

Figures 5 and 6 (page 53) show the fracture surfaces of cracks in the came Samples A and B, respectively. Both photographs were taken with the same settings under the same lighting conditions within minutes of each other. Note that the fracture surface of the Sample A crack is noticeably darker than that of the Sample B crack. A portion of the laboratory-created fracture is visible in Figure 6. This laboratorycreated crack is knife-edged and shiny. Contrast this bright, shiny appearance with the older fracture surfaces. The darker fracture surfaces are likely the result of greater oxidation. To verify that greater oxidation is the cause of the difference in appearance, these fracture surfaces were examined in a scanning electron microscope.

Scanning Electron Microscope (SEM) Analysis

Scanning electron microscope (SEM) analysis was used for two rea-

Table I Lead Samples for Analysis

| Sample | Date | Туре | |
|--------|----------|------------------|--|
| Λ | ~ 1913 | came and joint | |
| В | ~ 1930 | came | |
| С | new | came | |
| D | new | came | |
| Е | new | joint | |
| F | ~ 1970's | resoldered joint | |
| G | ~ 1970's | resoldered joint | |



Fig. 7: Sample A fracture. SEM shows ductile dimple rupture and wide patches of ductile tearing (straight arrow). Mag. 700X



Fig. 8: Sample A cracking (arrows) on flat surface proceeds into lead beyond oxide. Mag. 500X

sons in this investigation: to reveal crack fracture mode, if not too heavily corroded, and to analyze the surface for differing oxygen levels to see if there was a detectable difference between samples of different ages. A SEM is different from a binocular microscope in that it uses an electron beam instead of light to form an image of the surface being analyzed. This means that the resolution and depth-of-field is greatly increased. SEM analysis provides for viewing of



Fig. 9: Sample A cracking (arrows) on flat surface proceeds into lead beyond oxide. Mag. 250X



Fig. 10: Sample A large crack (arrow) in came near solder. Mag. 50X



Fig. 11: Location of cracks (arrow) on Sample A lead came. Mag. 15X

samples at much higher magnification than binocular microscopes.



Fig. 12: Sample A old fracture EDS spectrogram.



Fig. 14: Sample B old fracture EDS spectrogram.

The surfaces of the cracks shown in Figures 5 and 6 were examined in both the as-received and cleaned conditions. The oxide layer present on both surfaces obscured the fracture features in the as-received condition, so a light cleaning solution of a substance known as Alconox was used to remove it. After cleaning, both fracture surfaces exhibited ductile dimple rupture fracture mode with extensive stretching and tearing, Figure 7 (page 55). This indicates a very ductile, or deformable, metal. This is the same fracture mode that most ductile met-

als exhibit under tensile testing, except that the test specimens typically lack the tearing features. It represents exposure of the metal to a force beyond its physical capabilities to withstand. Such tearing could occur from unusually high wind gusts, undersized cames, lead creep, out-ofalignment panes, or the weight of the glass over time.

The flat surfaces of the Sample A came were also examined to check for cracks not visible to the naked eye. Several randomly selected regions were examined and approximately one



Fig. 13: Sample A new (laboratory-created) fracture EDS spectrogram.



Fig. 15: Sample B new (laboratory-created) fracture EDS spectrogram.

third of them possessed a crack. Several of these cracks are shown in Figures 8 through 10 (page 55). Note that the magnifications in these figures range from 50X to 500X. None of these cracks was visible to the naked eye, and only one was visible at 15X (shown in Figure 10 at 50X for greater clarity). Figure 11 (page 55) shows the region where the crack pictured in Figure 8 was located. Note that it is not visible at 15X. This means that any repairs carried out on visible cracks leave a multitude of cracks untouched and unremedied.

The oxide layer itself and any differences it might exhibit on cames of different ages were also examined. This examination occurred prior to cleaning. To analyze this, energy-dispersive x-ray spectroscopy, or EDS, was used to analyze the two fracture surfaces in question along with baseline laboratory-created fracture surfaces. EDS analysis uses equipment attached to a SEM to reveal the elements present in the analyzed region based upon characteristic x-ray emissions from the specimen. This is a qualitative microchemical analysis technique, meaning it detects relative amounts of elements. It cannot detect compounds (i.e., it will detect sodium and chlorine, but not sodium chloride) or determine percent composition. It will produce graphs, called spectrograms, that show peaks of various heights that correspond to an element's relative abundance in the analyzed region. In this way, it becomes easy to see in a graphic manner which region possesses more oxygen.

Figures 12 and 13 (page 56) are the spectrograms for Sample A old fracture (present when sample was received) and new fracture (laboratory-created). The difference in oxygen levels is readily apparent with the old fracture possessing an oxygen peak approximately three times as high as the laboratory-created fracture.

The difference is a little less striking in Figures 14 and 15 (page 56), which show the old and laboratory-created fracture oxygen levels of Sample B. The old fracture oxygen peak is approximately half again as high as the new fracture peak. Recall that Sample B is younger than Sample A, so age-related cracking would likely occur later in Sample B, assuming similarity of stresses and environment. This translates into less oxidation time for the Sample B crack than for the Sample A crack.

Oxidation produces a layer of corrosion product on the surface of a



An unsuccessful attempt to resolder old oxidized lead came.

crack. As time passes, this layer becomes thicker as more metal is consumed by the corrosion process. To evaluate the thickness of this layer, metallurgical mounts were created.

Metallurgical Analysis

Several samples were mounted in clear epoxy and ground and polished to reveal the interiors of soldered joints and profiles of cames. These resulting "mounts" were examined in the as-polished condition to provide for the best contrast between solder and came metal and any cracks, voids or inclusions present. Figure 16 (page 58) shows a new solder joint on new cames, Sample E, created for comparison. The cames joined by the solder are marked "C1" and "C2," and the solder is marked with an "S." Note that there are no gaps between the cames and the solder and the solder is solid with no inclusions (i.e., foreign particles), no cracks, no porosity (i.e., holes), or regions with lack of fusion. This was consistent along the entire joint.

Figure 17 shows a joint, Sample F, that was resoldered in the mid-1970s. Note the dark, round shapes indicative of porosity and how the new solder appears from the OD to join a much larger amount of metal than it actually does. At higher magnification, the extent of the lack of fusion is revealed to be even greater than it originally appeared in the lower magnification view, Figure 18. Large regions of porosity and lack of fusion such as this should not be present in a structural joint. The smooth profile of the new joint and solid fill of its solder provide a joint of greater soundness than the material of the resoldered joint. Porosity and lack of fusion represent discrete regions where gaps in the joint exist. The jagged profile of the joint creates sites known as "stress raisers," or places where the stresses the joint experiences are magnified due to geometry. Stress raisers can accelerate joint failure.

Metallurgical mounts also reveal the depth of any oxide layer present. Figures 19 through 21 show the pro-



Fig. 16: Sample E, new solder (S) joint on two new cames (C1 and C2). As-polished. Mag. 25X



Fig. 17: Sample F, old resoldered (1970s) joint showing solder (s) with porosity (curved clear arrow) and lack of fusion (curved solid arrows). Note that the lack of fusion extends to the small straight arrow above the two curved solid arrows. Mag. 18.75X



Fig. 18: Higher magnification view of Fig. 17 lack of fusion (arrows). As-polished. Mag. 100X



Fig. 19: New came (Sample C) in cross section showing no visible oxide layer. As-polished. Mag. 120X



Fig. 20: Oxide layer (arrows) on 1913 came (Sample A). As-polished. Mag. 150X



Fig. 21: Oxide layer remnant (arrows) on 1930 came (Sample B). As-polished. Mag. 150X

files of the came walls of the new sample, Sample C, and of older cames, Samples A and B, respectively. As expected, the new came, shown in Figure 19, possesses no visible oxide layer. Sample A, Figure 20, possesses a well-developed, tightly adhered oxide layer on the came OD. Debris visible on the came ID is caulking remnant. The oxide layer is approximately 0.008 inch thick. Lead is known to produce a protective oxide layer, so this very thin layer is expected and normal, even after approximately 91 years.

Sample B, dating from the 1930s, is shown in Figure 21. The oxide layer present on this sample is approximately 0.0005 inch thick. The thickness difference is negligible and the non-continuous layer of Sample B was very likely caused by oxide spalling, or falling off, during removal from its window.

In summary, metallurgical examination revealed negligible oxide-layer differences between the two older samples studied and a new sample. This is normal, as lead is known to produce an adherent, protective oxide layer when exposed to the elements. Once formed, a protective oxide layer greatly decreases further oxidation, and a relatively stable condition is achieved.

What this examination also revealed was a notable difference between a new joint and an older, resoldered joint. The new joint was solid, lacked porosity, and was well fused to the cames. The resoldered joint possessed porosity, lack of fusion, a jagged, stress-raising profile, and spotty fusion to the came. All these make the resoldered joint a much weaker construct.

The rationale behind resoldering old joints or only partially replacing cracked cames assumes the resulting joints are "good as new" if done "properly." Properly generally refers to

| | Table II Tensile Test Re | esults |
|------------------------|-----------------------------|--------------------------------|
| Tensile Test Sample | Age | Ultimate Tensile Strength, psi |
| Δ | ~ 1913 | 1,349 |
| В | new | 3,587 |
| С | ~ 1930 | 1,787 |
| D | new | 4,492 |

adequate cleaning, temperature control, flux selection and joint design.

However, as this and SEM examination showed, came cracks possess a layer of oxidation. No matter how well the flat came surface is scrubbed or cleaned, the crack fracture surface oxide layer, due to geometry, will persist. Fluxes are not substitutes for cleaning and cannot remove such persistent, well-adhered oxide layers. They should not be counted on to do so. Fluxes remove tarnish films from pre-cleaned surfaces, prevent oxidation during the soldering process, and lower the surface tension of the solder. Soldering over an oxide-filled crack will not produce a bond that is metallurgically equivalent to a new, uncracked length of came. It may even produce undesirable brittle intermetallic compounds in and near the solder joint that accelerate cracking of the joint.

As Figure 3 shows, cracking at resoldered joints is a concern. In addition to the crack, note the jagged came form and melt-through regions at this T-joint. These are all hallmarks of a poor bond. The melt-through and jagged eaten-away appearance of the came results from too high heat and/or too long a contact between the soldering tool and the came in these regions. All the stress-raiser issues previously discussed regarding uneven geometry are illustrated here. Cracks in the weld toe region, common in the samples examined from different windows, are the result of the metal attempting to accommodate strains induced by the soldering. This can be

due to excessive heat application, entrapped flux, creation of brittle intermetallic compounds, or poor stress distribution elsewhere along the came due to other repair work.

The prominence of such cracks in the samples examined from different sources suggests that they are less the result of the skill level of the person resoldering the joint (although the overall quality of the Figure 3 joint is very low) than of the difficulty in properly cleaning and designing a repair joint involving old, oxidized lead.

Also, as noted in the SEM examination section, the visible cracks are not the only cracks present on the came. Many of the cracks present on the came surfaces examined were visible only at magnifications over 100X. Even assuming that they could be resoldered properly, locating all such cracks on a sample intended for repair would require extensive microscopic examination.

Tensile Testing

Tensile testing was performed on samples of older cames and samples of new cames. Tensile testing was chosen as a test for this evaluation because it can provide an at-a-glance comparison between specimens. This type of testing pulls a specimen in tension at a slow, controlled rate until the specimen ruptures, or breaks. The sample cames, both old and new, were pulled in tension "as-is," or in their came configuration rather than as a machined tensile-test specimen.

This provided a real-world comparison between samples, as cracks present in the old cames were not eliminated by machining. The results of this testing are summarized in Table II. Note that the sample designations here are specific to this testing and do not refer to Table I sample designations. The new cames tested were chosen based upon size to compare with older cames. This means that one new Sample B was the same size and configuration as Sample A; and Sample D was the same size and configuration as Sample C. This is shown in Figure 22. These results indicate that the strength of a new came is a minimum of two and a half times that of an old came. In other words, using a new came provides 250% more tensile strength than the old cames. Since the lead cames are the structural framework for the glass, this translates into a much greater ability to withstand the weight of the glass and the wind loads to which windows are subjected. This is significant because SEM examination of an older crack fracture surface showed a fracture mode consistent with an overload failure, the same type of failure a tensile test produces.

The practice of allowing a buckled window to settle and pressing it flat again will not heal the cracks that were instrumental in producing the lowered tensile strength of the two older cames. In fact, attempting to press buckled and distorted came walls back into position can extend cracks already present, as well as cause new ones, when the stretched metal is forced to lie flat again. This is a simple geometric response. The lead came walls cannot "unstretch."

Chemical Analysis

Chemical analysis was performed on came Samples A, B, and C to determine if any compositional differences existed between the older leads from



Figure 22: Old and new came samples for tensile testing.

| | Table III Chemical Analysis Results | | | | | |
|-----------|--|----------------|---------------|--|--|--|
| Element | Composition, weight % | | | | | |
| | Sample A, 1913 | Sample B, 1930 | Sample C, new | | | |
| Antimony | 0.12 | 0.14 | 0.78 | | | |
| Arsenic | < 0.0002 | < 0.0002 | 0.001 | | | |
| Bismuth | 0.080 | 0.025 | 0.018 | | | |
| Calcium | < 0.0002 | < 0.0002 | < 0.0002 | | | |
| Copper | 0.004 | 0.033 | 0.027 | | | |
| Iron | < 0.0002 | < 0.0002 | < 0.0002 | | | |
| Lead | Remainder | Remainder | Remainder | | | |
| Lithium | < 0.0002 | < 0.0002 | < 0.0002 | | | |
| Nickel | < 0.0002 | < 0.0002 | < 0.0002 | | | |
| Silver | 0.004 | 0.005 | 0.006 | | | |
| Sulfur | < 0.001 | < 0.001 | < 0.001 | | | |
| Tellurium | < 0.0002 | < 0.0002 | 0.0002 | | | |
| Tin | 0.031 | 0.064 | 0.26 | | | |
| Zinc | 0.0004 | 0.0003 | 0.0005 | | | |

the early twentieth century and new lead ordered to "restoration quality." The results are summarized in Table III. These analysis results indicate that the lead cames from 1913 and 1930 (Samples A and B, respectively) are very similar to each other and are also similar to two Unified Numbering System alloys: L52505 Lead-Antimony alloy and L52510, 99.8% Lead. This is consistent with manufacturing efforts of the time to produce high-purity lead for window cames.

The new "restoration lead" (Sample C) contains a much higher level of antimony and tin than the

older lead. This alloy is similar to many UNS alloys, among them: L52560 Bullet Alloy, L52615 Lead-Base Die Casting Alloy, etc.

The new lead contains a larger amount of elements known to produce something known as solid-solution hardening effects (i.e., antimony, bismuth, arsenic, tin, etc.). This means that lead with the chemical composition of the new lead would be slightly stronger than lead with the chemical composition of the old lead, even if both samples were in a new, uncracked condition. A stronger alloy is capable of withstanding service conditions better than a weaker alloy.

Conclusion

Several conclusions can be drawn from the analysis results and review of repair and releading techniques. These are presented below as a bullet list. For greater detail, refer to the *History and Discussion* section as well as individual testing results.

- Modern refining techniques produced lead of much greater purity for use in mid-ninteenth century to mid-twentieth century windows.
 This lead is very different from both medieval lead and its modern restoration-lead counterpart.
- Lead of greater purity is a weaker metal than alloyed medieval lead and modern restoration lead. As a result, the pure lead is less able to withstand glass weight and wind loads than its alloyed relatives. Came-wall stretching and cracking will eventually result.
- Pressing a buckled window flat does not repair cracks in the cames. The pressing process is likely to propagate existing cracks and create new ones.
- Resoldering old joints in old cames results in poor joint quality and can induce further cracking at the solder pool toe. This does not restore the window lead framework to "good as new" condition.
- Window buckling due to lead framework structural failure transfers loads that were previously handled by the lead to the glass panes. This is a recipe for glass breakage due to its inherent brittleness.
- Modern "restoration quality" lead came consists of an alloy based upon chemical analysis of some medieval leads. Use of this alloyed lead in restoration of windows should result in the greater ability of the restored lead framework to withstand service loads over the purer lead used in the late ninteenth and early twentieth centuries. However, as with all structural frameworks, even the restoration lead will eventually require replacement.
- The cracks in cames visible to the naked eye are not the only cracks

present. Soldering over visible cracks does not eliminate microscopic cracks. Cracking weakens cames and reduces their ability to withstand service loads.

• Tensile-strength testing revealed new-came strength to be a minimum of 250% higher than crackedcame strength.

"...pressing a buckled window flat will not heal the cracks...and in fact may cause new ones when the stretched metal is forced to lie flat again. This is a simple geometric response. The lead came walls cannot unstretch."

• Came cracking is an inevitable result of service due to the inherent ability of lead to creep at normal use temperatures and to resist heat treatment and work-hardening procedures used regularly with other alloys. While the solid-solution strengthening made possible with the use of certain alloys makes stronger cames available, even these will eventually experience structural failure due to the intrinsic behavior of their lead base.

Finally, this analysis shows that the service life of the lead-support system in a stained glass window is influenced by more than age. While the testing indicates that the older the lead is, the greater the likelihood of failure, chemical composition certainly influences its life span.

Traditionally, we may think that lead should be near 100 years old before considering replacement, but if it is relatively pure, that time span may be greatly shortened. Taking into account many other factors, including wind loads, climate ranges, installation types and design style, the correct response to the signs of structurally failing lead is complete replacement with a new lead matrix.

About the Author

Veda-Anne Ulcickas is a Materials Engineer in the Failure Analysis/Materials Engineering Consulting Dept. at Massachusetts Materials Research, Inc. (MMR). She holds a BS in Mechanical Engineering and an MS in Materials Science and Engineering, both from Worcester Polytechnic Institute. Her graduate work investigated the suitability of using fractal-based advanced computational methods as a quantitative surface-measurement tool.

Ms. Ulcickas' work at MMR has included product-design assistance for sporting goods and housewares, material consultations for water treatment plants and other severe-service environments, as well as a wide variety of failure investigations. Her services are regularly requested for investigations into natural gas fires and explosions, water main breaks, and aircraft-industry product failures and quality non-compliances. She has extensive experience evaluating brazed and soldered joints for such diverse industries as electronics, rock cutting, and aerospace component manufacturing.

Massachusetts Materials Research, Inc. specializes in the practical application of advanced testing, engineering, inspection and failure-analysis technology. They have recently been involved in investigating the failures of various aircraft engine components, inspection of observatory azimuth gears and large, mobile tractor-trailer x-ray units, and investigating several fires and explosions involving household appliances and utilities. MMR also conducted the inspection of infrastructure used to produce the Jeep™ Parking Only commercial in New York City, which can be viewed on the jeep.com website.

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